CATHERINE E. HOUSECROFT & ALAN G. SHARPE INORGANIC CHERNERAL STREAM



Periodic table

Č	1	ľ	ž.		Atomic number, Z	N							ç
Ē	₩ 		<u>_</u>	ement :	Element symbol								18
- Relå	▲ Relà	Rela	e e	ative a	Relative atomic mass, A _r	iass, A _r		13	14	15	16	17	He ²
	1							ъ	9	7	8	6	10
								۵	υ	z	0	ш	Ne
								10.81	12.01	14.01	16.00	19.00	20.18
								13	14	15	16	17	18
6	8 8		6		10	11	12	AI 26.98	Si 28.09	Р 30.97	S 32.06	C 35.45	Ar 39.95
27			27		28	29	30	31	32	33	34	35	36
ပိ			ပိ		Ż	Cu	Zn	Ga	ge	As	Se	Br	Ϋ́
58.93			58.93		58.69	63.54	65.38	69.72	72.63	74.92	78.97	79.91	83.80
45			45		46	47	48	49	50	51	52	53	54
Rh	Ru Rh		Rh		Pd	Ag	B	<u>_</u>	Sn	Sb	Те		Xe
102.91			02.91	_	106.42	107.87	112.41	114.82	118.71	121.76	127.60	126.90	131.29
77			77		78	79	80	81	82	83	84	85	86
<u> </u>			<u> </u>		Ł	Ν	Hg	F	Pb	Bi	Ро	At	Rn
192.22	_	_	92.23	~	195.08	196.97	200.59	204.38	207.2	208.98	[209]	[210]	[222]
109			109		110	111	112	113	114	115	116	117	118
Ę			Ę		Ds	Rg	5	ЧN	ᇿ	Ĕ	2	Ts	0g
[278]			[278]		[281]	[281]	[285]	[285]	[289]	[289]	[293]	[294]	[294]

	57	58	59	60	61	62		64	65	66	67	68	69	70	71
Lanthanoids	La	e U	Pr	PN	Pm	Sm	Eu	рд	Tb	D	Р	ш	Tm	٩۲	Lu
	138.91	138.91 140.12 140.91	140.91	144.24	[145]	150.36	•	157.25	158.93	162.50	164.93	167.26	168.93	173.04	174.97
	89	06	91	92	93	94		96	97	98	66	100	101	102	103
Actinoids	Υc	Ч	Pa	C	Np	Pu		۳ د	Bk	უ	Es	Fm	ΡM	No No	Ľ
	[227]	[227] 232.04 231.04	231.04	238.03	[237]	[244]		[247]	[247]	[251]	[252]	[257]	[258]	[259]	[262]
Mass numbers in [] refer to elements which possess no stable nuclides; Pa and U, the value of A_r is based on the terrestrial isotopic compositio	to elemen s based oi	ts which _I n the terr	possess no estrial iso	o stable n topic com	uclides; t iposition	he mass I	number g	stable nuclides; the mass number given is for the longest-lived isotope of the element. For each of Th, opic composition.	r the lon	gest-livec	isotope	of the el	ement. Fo	or each of	Th,

APPLICATIONS

Box 11.2 Keeping time with caesium

In 1993, the National Institute of Standards and Technology (NIST) brought into use a caesium-based atomic clock called NIST-7 which kept international standard time to within one second in 10^6 years. The system depends upon repeated transitions from the ground to a specific excited state of atomic Cs, and the monitoring of the frequency of the electromagnetic radiation emitted.

In 1995, the first caesium fountain atomic clock was constructed at the Paris Observatory in France. A fountain clock, NIST-F1, was introduced in 1999 in the US to function as the country's primary time and frequency standard. In 2014, NIST-F1 was superseded by NIST-F2 which is accurate to within one second in 300×10^6 years, approximately three times more accurate than NIST-F1. In the UK, the National Physics Laboratory's caesium fountain atomic clock NPL-CsF2 (shown in the photograph) is accurate to within one second in 138×10^6 years. While earlier caesium clocks observed Cs atoms at ambient temperatures, caesium fountain clocks use lasers to slow down and cool the atoms to temperatures approaching 0K. Current atomic clock research is focusing on instruments based on optical transitions of neutral atoms or of a single ion (e.g. ⁸⁸Sr⁺). Progress in this area became viable after 1999 when optical counters based on femtosecond lasers (see Box 26.2) became available.

Extremely accurate time-keeping is fundamental to many technological advances such as global positional system (GPS) satellite receivers. Mobile phones and laptops with built-in clocks (which we assume are accurate!) are part of our everyday lives.

Further reading

- M. Chalmers (2009) *New Scientist*, vol. 201, issue 2694, p. 39– 'Every second counts'.
- T.P. Heavner *et al.* (2014) *Metrologia*, vol. 51, p. 174 'First accuracy evaluation of NIST-F2'.

Self-study exercises

- 1. Why is it not necessary to include the 1s, 2s and 2p orbitals and electrons in the MO description of the bonding in Na₂?
- 2. Use the MO diagram for Na₂ to determine whether Na₂ is paramagnetic or diamagnetic. [*Ans*: Diamagnetic]

See end-of-chapter problem 11.5 for an extension of these exercises.

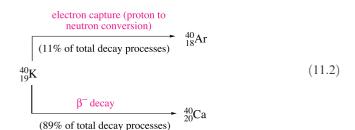
Radioactive isotopes

In addition to the radioactivity of Fr, 0.02% of naturally occurring K consists of 40 K which decays according to scheme 11.2.



The caesium fountain atomic clock NPL-CsF2 in the National Physics Laboratory in the UK.

- R. Le Targat *et al.* (2013) *Nature Commun.*, vol. 4, p. 2109 'Experimental realization of an optical second with strontium lattice clocks'.
- R. Li, K. Gibble and K. Szymaniec (2011) *Metrologia*, vol. 48, p. 283 – 'Improved accuracy of the NPL-CsF2 primary frequency standard: evaluation of distributed cavity phase and microwave lensing frequency shifts'.
- Y. Ovchinnikov and G. Marra (2011) *Metrologia*, vol. 48, p. 87– 'Accurate rubidium atomic fountain frequency standard'.
- M. Takamoto, F.-L. Hong, R. Higashi and H. Katori (2005) *Nature*, vol. 435, p. 321 'An optical lattice clock'.
- R. Wynands and S. Weyers (2005) *Metrologia*, vol. 42, p. S64 'Atomic fountain clocks'.
- www.nist.gov/pml/time-and-frequency-division/primary-standardnist-f1/



The overall half-life for both the β -decay and electron capture is 1.25×10^9 yr.

The decay of ⁴⁰K provides the human body with a natural source of radioactivity, albeit at very low levels. The decay from ⁴⁰K to ⁴⁰Ar is the basis of a technique for dating minerals (e.g. biotite, hornblende and volcanic rocks). When volcanic magma cools, ⁴⁰Ar formed from the decay of ⁴⁰K remains trapped in the mineral. Crushing and heating rock samples releases argon, and the amount of ⁴⁰Ar present

APPLICATIONS

Box 11.3 Alkali metal ion batteries

The sodium/sulfur battery operates around 570-620 K and consists of a molten sodium anode and a liquid sulfur cathode which contains a carbon fibre matrix for conduction. The anode and cathode are separated by a solid β -alumina electrolyte (see Section 28.2). The cell reaction is:

$$2Na(l) + nS(l) \longrightarrow Na_2S_n(l)$$
 $E_{cell} = 2.08 V$

and this is reversed when the battery is recharged by changing the polarity of the cell. In the 1990s, it appeared that sodium/ sulfur batteries may have potential application in the electric vehicle (EV) market, but the high operating temperature of the sodium/sulfur battery is a drawback to the motor industry, and other battery technologies have superseded these batteries for electric and hybrid electric vehicles. Stationary sodium/ sulfur batteries are used for energy storage, notably in Japan. This application follows from the fact that self-discharge from sodium/sulfur batteries occurs only at very low levels.

An important advance in battery technology has been the development of rechargeable, high energy-density lithium-ion (Li-ion) batteries, first introduced to the commercial market in 1991. Since then, the market has grown enormously; in 2015, 35% of Li metal produced globally ended up in rechargeable Li-ion batteries. The Li-ion battery has a cell potential of 3.6 V and consists of a positive LiCoO₂ electrode separated from a graphite electrode by a solid electrolyte across which Li⁺ ions can migrate when the cell is charging. In commercial Li-ion batteries, the electrolyte is usually LiPF₆ in a solution of two or more alkylcarbonates. Li-ion batteries are manufactured in a discharged state. Solid LiCoO2 adopts an α-NaFeO2 structure type in which the O atoms are approximately cubic close-packed. The octahedral holes are occupied by M(I) or M'(III) (Li^+ or Co^{3+} in LiCoO₂) in such a way that the different metal ions are arranged in layers. During charging, Li⁺ ions move out of these layers, are transported across the electrolyte, and are intercalated by the graphite (see Section 14.4). During discharge of the cell, the Li⁺ ions return to the metal oxide lattice. The cell reaction can be represented as follows:

$$LiCoO_2 + C_6(graphite) \xrightarrow{charge} Li_{1-x}CoO_2$$

 $+ Li_x C_6$ (intercalated graphite)

The cobalt centres are redox active, some being oxidized from Co(III) to Co(IV) as Li^+ is removed from $LiCoO_2$. The crucial factor in lithium-ion batteries is that both electrodes are able to act as hosts for Li^+ ions. Rechargeable, Li-ion batteries now dominate the market for small electronic devices such as laptop computers, iPads, mobile phones, iPods and MP3 players, and in electric bicycles and cordless power tools.

A disadvantage of Li-ion batteries containing cobalt is their relatively high cost. Current research strategies are aimed at finding replacement electrode materials both to increase battery performance and to reduce cost. Two contenders are LiMn₂O₄ and LiFePO₄. LiMn₂O₄ has a spinel structure (see Box 13.7) and when coupled with a graphite electrode forms a Li-ion battery, the cell reaction of which is summarized below:

$$LiMn_2O_4 + C_6(graphite) \xrightarrow{charge} Li_{1-x}Mn_2O_4$$

 $+ Li_x C_6$ (intercalated graphite)

Applications of this type of Li-ion battery include those in hybrid electric vehicles (HEVs). Manufacturers including



A lithium-ion battery pack for General Motors' electric vehicle, the Chevrolet Volt, which has been on the market since 2011.

Toyota and Honda produce hybrid electric and plug-in electric vehicles (rechargeable from an external power supply when the car is parked) incorporating Li-ion batteries, but the first massproduced HEV containing a Li-ion battery was launched by Mercedes-Benz in 2009. In the S400 Blue Hybrid, a 120 V Li-ion battery pack powers an electric motor which works in conjunction with an internal combustion engine, the operating mode being computer controlled. A regenerative braking system (see Box 10.5) converts kinetic energy to electrical energy which is stored in the battery, and the electrical motor also recovers energy during deceleration.

Graphite anodes currently employed in commercial Li-ion batteries have a low theoretical charge capacity (372 mAh g^{-1}) . This must be improved for commercial use of Li-ion batteries in electric vehicles and other high-energy storage applications. Cutting-edge research is being carried out into the replacement of the intercalation reactions discussed above by alloying reactions. For example, Li reacts with Si to give alloys such as $\text{Li}_{15}\text{Si}_4$ and $\text{Li}_{22}\text{Si}_5$, and the theoretical charge capacity of Si is 3579 mAh g^{-1} . Despite this high capacity, bulk Si suffers from a large variation in volume during battery charging and discharging. A viable compromise is the use of C/Si composites, for example, the addition of Si to graphite produces anodes with greatly improved capacity.

Further reading on lithium-ion batteries

- L. Liu, J. Lyu, T. Li and T. Zhao (2016) *Nanoscale*, vol. 8, p. 701 'Well-constructed silicon-based materials as high-performance lithium-ion battery anodes'.
- D. Ma, Z. Cao and A. Hu (2014) *Nano-Micro Lett.*, vol. 6, p. 347 – 'Si-Based anode materials for Li-ion batteries: A mini review'.
- N. Nitta and G. Yushin (2014) Part. Part. Syst. Charact., vol. 31, p. 317 – 'High-capacity anode materials for lithium-ion batteries: Choice of elements and structures for active particles'.
- C.-M. Park, J.-H. Kim, H. Kim and H.-J. Sohn (2010) *Chem. Soc. Rev.*, vol. 39, p. 3115.
- B. Scrosati and J. Garche (2010) *J. Power Sources*, vol. 195, p. 2419.
- F.T. Wagner, B. Lakshmanan and M.F. Mathias (2010) J. Phys. Chem. Lett., vol. 1, p. 2204.

can be determined by mass spectrometry. Atomic absorption spectroscopy is used to determine the 40 K content. The age of the mineral can be estimated from the ratio of 40 K : 40 Ar.[†]

NMR active nuclei

Each of the alkali metals has at least one NMR active nucleus (Table 11.1), although not all nuclei are of sufficient sensitivity to permit their routine use. For examples of NMR spectroscopy utilizing *s*-block metals, see Section 4.8 and worked example 23.1.

11.4 The metals

Appearance

The metals Li, Na, K and Rb are silvery-white, but Cs has a golden-yellow cast. All are soft, Li the least so, and the trend is consistent with their melting points (Table 11.1). The particularly low melting point of Cs (301.5 K) means that it may be a liquid at ambient temperatures in hot climates.

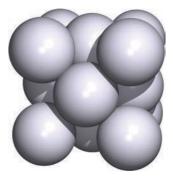
Reactivity

We have already described the behaviour of the metals in liquid NH_3 (see Section 9.6). The ultimate products are alkali metal amides, MNH_2 (see eq. 9.28), and $LiNH_2$, $NaNH_2$ and KNH_2 are important reagents in organic synthesis. In the solid state, these amides adopt structures consisting of cubic close-packed $[NH_2]^-$ ions with M^+ ions occupying half the tetrahedral holes.

Worked example 11.2 Structure of NaNH₂

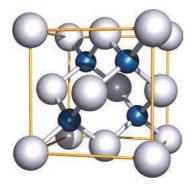
The solid state structure of $NaNH_2$ can be approximately described as consisting of an fcc arrangement of amide ions with Na^+ ions occupying half the tetrahedral holes. To which structure type (or prototype structure) does this correspond?

A face-centred cubic (i.e. cubic close-packed) arrangement of $[NH_2]^-$ ions (assuming each is spherical) corresponds to the following unit cell:



 † For a discussion of $^{40}\mathrm{K-}^{40}\mathrm{Ar}$ dating, see: W.A. Howard (2005) J. Chem. Educ., vol. 82, p. 1094.

There are eight tetrahedral holes within the unit cell. The Na^+ ions occupy half of these interstitial sites:



 $NaNH_2$ adopts a zinc blende (ZnS) structure (compare with Fig. 6.21b).

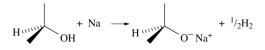
Self-study exercises

- 1. Use the diagram of the unit cell for sodium amide to confirm the $1:1 \text{ Na}^+:[\text{NH}_2]^-$ ratio.
- Using the diagram of the unit cell of NaNH₂, determine the coordination number of each [NH₂]⁻ ion. To check your answer, think how this coordination number must be related to that of an Na⁺ ion.

Although Li, Na and K are stored under a hydrocarbon solvent to prevent reaction with atmospheric O_2 and water vapour, they can be handled in air, provided undue exposure is avoided; Rb and Cs should be handled in an inert atmosphere. Lithium reacts quickly with water (eq. 11.3); Na reacts vigorously, and K, Rb and Cs react violently with the ignition of H₂ produced.

$$2Li + 2H_2O \longrightarrow 2LiOH + H_2 \tag{11.3}$$

Sodium is commonly used as a drying agent for hydrocarbon and ether solvents. Sodium should *never* be used to dry halogenated solvents (see eq. 14.47). The disposal of excess Na must be carried out with care and usually involves the reaction of Na with propan-2-ol:



This is a less vigorous, and therefore safer, reaction than that of Na with H_2O or a low molecular mass alcohol. An alternative method for disposing of small amounts of Na involves adding H_2O to a sand-filled ceramic container (e.g. plant pot) in which the metal has been buried. The

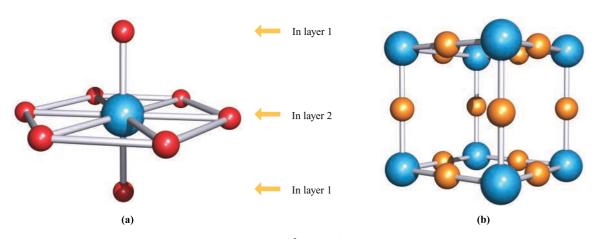


Fig. 11.4. (a) The solid state structure of Li_3N consists of layers of N^{3-} and Li^+ ions (ratio 1:2) alternating with layers of Li^+ ions; the latter are arranged such that they lie over the N^{3-} ions. Each N centre is in a hexagonal bipyramidal (8-coordinate) environment; there are two types of Li^+ ion, those in layer 1 are 2-coordinate, and those in layer 2 are 3-coordinate with respect to the N centres (see end-of-chapter problem 11.12). (b) The unit cell of sodium nitride; Na₃N adopts an anti-ReO₃ structure. Colour code: N, blue; Li, red; Na, orange.

conversion of Na to NaOH occurs slowly, and the NaOH reacts with the sand (i.e. SiO_2) to yield sodium silicate.[†]

All the group 1 metals react with the halogens (eq. 11.4) and H_2 when heated (eq. 11.5). The energetics of metal hydride formation are essentially like those of metal halide formation, being expressed in terms of a Born–Haber cycle (see Section 6.14).

 $2M + X_2 \rightarrow 2MX$ X = halogen (11.4)

 $2M + H_2 \longrightarrow 2MH \tag{11.5}$

Self-study exercises

1. Calculate $\Delta_{\text{lattice}} H^{\text{o}}(\text{NaH, s})$ using a Born–Haber cycle; $\Delta_{\text{f}} H^{\text{o}}(\text{NaH, s}) = -56.3 \text{ kJ mol}^{-1}$, and see the Appendices for other data.

 $[Ans. -805.1 \text{ kJ mol}^{-1}]$

2. Use the VBT approach (Section 6.16) to estimate a value for the lattice energy of NaH.

 $[Ans. -807 \,\mathrm{kJ}\,\mathrm{mol}^{-1}]$

Lithium reacts spontaneously with N_2 , and reaction 11.6 occurs at 298 K to give red-brown, moisture-sensitive lithium nitride. Solid Li₃N has an interesting structure (Fig. 11.4a) and a high ionic conductivity (see Section 28.2). Attempts to prepare the binary nitrides of the later alkali metals were not successful until 2002. Na₃N (which is very moisture-sensitive) may be synthesized in a vacuum chamber by depositing atomic sodium and nitrogen onto

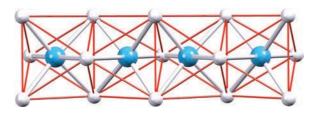


Fig. 11.5. Part of one column of face-sharing octahedra (shown in red) of K^+ ions in the hexagonal phase of K_3N . This is an anti-TiI₃ structure-type. Colour code: K, pale grey; N, blue.

a cooled sapphire substrate and then heating to room temperature. The structure of Na_3N is very different from that of Li_3N (Fig. 11.4), with Na_3N adopting an anti-ReO₃ structure (see Fig. 22.4 for ReO₃) in which the Na^+ ions are 2-coordinate and the N^{3-} ions are octahedrally sited.

$$6Li + N_2 \longrightarrow 2Li_3N \tag{11.6}$$

An analogous synthetic method to that used to prepare Na_3N was reported in 2004 for the synthesis of K_3N from potassium and nitrogen.[‡] Two phases of K_3N have been identified. The hexagonal phase adopts an anti-TiI₃ structure; the K^+ ions lie at the vertices of face-sharing octahedra (Fig. 11.5), forming columns which lie parallel to one another in the lattice. The coordination numbers of the K^+ and N^{3-} ions are 2 and 6, respectively. Reactions of the alkali metals with O_2 are discussed in Section 11.6.

Acetylides, M_2C_2 , are formed when Li or Na is heated with carbon. These compounds can also be prepared by treating the metal with C_2H_2 in liquid NH₃. Reactions

[†] See: H.W. Roesky (2001) *Inorg. Chem.*, vol. 40, p. 6855 – 'A facile and environmentally friendly disposal of sodium and potassium with water'.

[‡]D. Fischer, Z. Cancarevic, J.C. Schön and M. Jansen (2004) Z. Anorg. Allg. Chem., vol. 630, p. 156.

between K, Rb or Cs and graphite lead to a series of intercalation compounds MC_n (n = 8, 24, 36, 48 and 60) in which the alkali metal atoms are inserted between the layers in a graphite host lattice (see structure 14.2 and Fig. 14.4a). For a given formula, the compounds are structurally similar and exhibit similar properties, irrespective of the metal. Under high-pressure conditions, MC_{4-6} (M = K, Rb, Cs) can be formed. In contrast, the intercalation of lithium into graphite gives LiC_6 , LiC_{12} , LiC_{18} and LiC_{27} . At high pressures, LiC₂₋₄ can be produced. In contrast to the other alkali metals, sodium does not intercalate into graphite. However, in 2014, it was discovered that solvated Na⁺ ion intercalation can occur. This may lead towards applications of Na/solvent co-intercalation anode materials for rechargeable batteries.[†] We return to graphite intercalation compounds in Section 14.4. Note that, in contrast to the above intercalations, intercalation in Li-ion batteries (Box 11.3) does not involve Li/Li⁺ redox chemistry.

The alkali metals dissolve in Hg to give amalgams (see Box 22.2). Sodium amalgam is a liquid only when the percentage of Na is low. It is a useful reducing agent in inorganic and organic chemistry, and can be used in aqueous media because there is a large overpotential for the discharge of H_2 .

An innovative method of handling alkali metals is to absorb them into silica gel, thus providing a convenient source of the metals as powerful reducing agents, e.g. in Birch reductions:



 $Na_2K-SG = Na_2K$ alloy in silica gel

Foreseeable applications of these materials are in the use of continuous-flow columns for reduction reactions in, for example, the pharmaceutical industry. The silica gel–alkali metal powders react quantitatively with water, liberating H₂. Since the powders are easily handled and stored, they have the potential to act as a 'supply-on-demand' source of H₂,[‡] and are now available commercially.

11.5 Halides

The MX halides (see Chapter 6 for structures) are prepared by direct combination of the elements (eq. 11.4) and all the halides have large negative $\Delta_f H^o$ values. However, Table 11.2 shows that for X = F, values of $\Delta_f H^o(MX)$ become *less negative* down the group, while the reverse trend is true for X = Cl, Br and I. For a given metal, $\Delta_f H^o(MX)$ always becomes less negative on going from MF to MI. These generalizations can be explained in terms of a Born–Haber cycle. Consider the formation of MX (eq. 11.7) and refer to Fig. 6.27.

$$\Delta_{f}H^{o}(MX, s) = \underbrace{\{\Delta_{a}H^{o}(M, s) + IE_{1}(M, g)\}}_{\uparrow} + \underbrace{\{\Delta_{a}H^{o}(X, g) + \Delta_{EA}H(X, g)\}}_{\uparrow} + \underbrace{\{\Delta_{a}H^{o}(X, g) + \Delta_{EA}H(X, g)\}}_{halide-dependent term} + \Delta_{lattice}H^{o}(MX, s)$$
(11.7)

For MF, the variable quantities are $\Delta_a H^o(M)$, $IE_1(M)$ and $\Delta_{\text{lattice}} H^{\circ}(\text{MF})$, and similarly for each of MCl, MBr and MI. The sum of $\Delta_a H^o(M)$ and $IE_1(M)$ gives for the formation of Li⁺ 681, of Na⁺ 604, of K⁺ 509, of Rb⁺ 485 and of Cs^+ 454 kJ mol⁻¹. For the fluorides, the trend in the values of $\Delta_{\rm f} H^{\rm o}({\rm MF})$ depends on the relative values of $\{\Delta_a H^o(\mathbf{M}) + IE_1(\mathbf{M})\}\$ and $\Delta_{\text{lattice}} H^o(\mathbf{MF})$ (Table 11.2), and similarly for chlorides, bromides and iodides. Inspection of the data shows that the variation in $\{\Delta_a H^o(M) + IE_1(M)\}$ is *less* than the variation in $\Delta_{\text{lattice}} H^{\circ}(\text{MF})$, but greater than the variation in $\Delta_{\text{lattice}} H^{\text{o}}(\text{MX})$ for X = Cl, Br and I. This is because lattice energy is proportional to $1/(r_+ + r_-)$ (see Section 6.13) and so variation in $\Delta_{\text{lattice}} H^{\circ}(MX)$ for a given halide is greatest when r_{-} is smallest (for F^{-}) and least when r_{-} is largest (for I⁻). Considering the halides of a given metal (eq. 11.7), the small change in the term $\{\Delta_{a}H^{o}(X) + \Delta_{EA}H(X)\} (-249, -228, -213, -188 \text{ kJ mol}^{-1})$ for F, Cl, Br, I respectively) is outweighed by the decrease in $\Delta_{\text{lattice}} H^{\circ}(\text{MX})$. In Table 11.2, note that the *difference* between the values of $\Delta_{\rm f} H^{\rm o}({\rm MF})$ and $\Delta_{\rm f} H^{\rm o}({\rm MI})$ decreases significantly as the size of the M⁺ ion *increases*.

The solubilities of the alkali metal halides in water are determined by a delicate balance between lattice energies and Gibbs energies of hydration (see Section 7.9 for $\Delta_{sol}G^{o}$ and $\Delta_{hyd}G^{o}$). LiF has the highest lattice energy of the group 1 metal halides and is only sparingly soluble, but solubility relationships among the other halides call for detailed discussion beyond the scope of this book.^{*} The salts LiCl, LiBr, LiI and NaI are soluble in some *O*-donor organic solvents, e.g. LiCl dissolves in THF and MeOH. Complexation of the Li⁺ or Na⁺ ion by the *O*-donors is likely in all cases (see Section 11.8). Both LiI and NaI are very soluble in liquid NH₃, forming complexes; the unstable complex [Na(NH₃)₄]I has been isolated and contains a tetrahedrally coordinated Na⁺ ion.

In the vapour state, alkali metal halides are present mainly as ion-pairs, but measurements of M-X bond distances and

[†] See: H. Kim *et al.* (2015) *Energy Environ. Sci.*, vol. 8, p. 2963 – 'Sodium intercalation chemistry in graphite'.

[‡] See: J.L. Dye *et al.* (2005) *J. Am. Chem. Soc.*, vol. 127, p. 9338; M. Shatnawi *et al.* (2007) *J. Am. Chem. Soc.*, vol. 129, p. 1386.

^{*} For further discussion, see: W.E. Dasent (1984) *Inorganic Energetics*, 2nd edn, Cambridge University Press, Cambridge, Chapter 5.